## Diagrammatic Separation of Different Crystal Structures of A<sub>2</sub>BX<sub>4</sub> Compounds Without Energy Minimization: A Pseudopotential Orbital Radii Approach

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<sub>2</sub>BX<sub>4</sub> family of compounds manifest a wide range of physical properties, including transparent conductivity, ferromagnetism, and superconductivity. A 98% successful diagrammatic separation of the 44 different crystal structures of 688 oxide A<sub>2</sub>BX<sub>4</sub> compounds (96% for 266 oxide-only) is described by plotting the total radius of the A atom*R*<sub>A</sub> versus the radius of the B atom *R*<sub>B</sub> for many A<sub>2</sub>BX<sub>4</sub> compounds of known structure types and seeking heuristically simple, straight boundaries in the *R*<sub>A</sub> versus *R*<sub>B</sub> plane that best separate the domains of different structure types. The radii are sums *R*<sub>A</sub> = *R*<sub>s</sub>(A) + *R*<sub>p</sub>(A) of the quantum-mechanically calculated ''orbital radii'' *R*<sub>s</sub>(*R*<sub>p</sub>), rather than empirical radii or phenomenological electronegativity scales. These success rates using first-principles orbital radii uniformly exceed the success rates using classic radii. Such maps afford a quick guess of the crystal structure of a yet unmade A<sub>2</sub>BX<sub>4</sub> compound by placing its atomic orbital radii on such maps and reading off its structure type.

## 1. Introduction

A<sub>2</sub>BX<sub>4</sub> compounds<sup>[1–6]</sup> constitute a centrally important group in inorganic solid state, manifesting a wide range of physical phenomena including insulation, transparent conductivity, ferromagnetism, ferroelectricity, and superconductivity. The 790 known  $A_2BX_4$  compounds<sup>[1-13]</sup> are distributed into 44 different crystal structure types as listed in Tables 1 and 2 (the Supporting Information lists the compounds belonging to each of the different structure types). This group of compounds exhibits significant chemical diversity, including chalcogen anions (X = O, S, Se, Te)as well as halides (X = F, Cl, Br, I), nitrides (X = N), cyanides (X = CN), and even nitrites  $(X = NO_2)$ . The cations manifest cases where both A and B are main-group metals (A and B = Al, Mg, Ge, Sn) or cases where both A and B are transition elements (e.g., Ni<sub>2</sub>TiO<sub>4</sub>, V<sub>2</sub>MnO<sub>4</sub>) or cases where we have one of each (e.g.,  $Al_2NiO_4$ ) as well as rare-earth cationic species (e.g.,  $Yb_2FeS_4$ ). These compounds include cation-deficient structures (e.g., Adeficient A3B2X8 or B-dificient A4BX8) as well as cases such as  $A_3B_2X_6$  with cations in excess. The structures adopted by this group are no less fascinating than their chemical constitution. This family of compounds includes, for example, the spinel structure-type (255 members), the Th<sub>3</sub>P<sub>4</sub> structure-type (87 members), the Fe<sub>2</sub>CaO<sub>4</sub> structure-type (68 members), the K<sub>2</sub>SO<sub>4</sub> structure-type (69 members), the Cr<sub>3</sub>S<sub>4</sub> structure-type (57 members), and the Olivine (Al<sub>2</sub>BeO<sub>4</sub>) structure-type (48 members).<sup>[14]</sup> The A<sub>2</sub>BX<sub>4</sub> structure-types differ in crystal classes (cubic, orthorhombic, rhombohedral) and local environments ("motifs"), covering tetrahedral and octahedral as well as 5- and 7-fold coordination sites.

An outstanding challenge in structural inorganic chemistry<sup>[2–4,6,15]</sup> and in solid-state physics<sup>[16,17]</sup> has been to explain the distribution of the known  $A_2BX_4$  compounds into different structure types. Two leading types of

approaches of predicting or rationalizing the crystal structure of a given  $A_2BX_4$  compound have developed. In the deductive approach, one explicitly varies the structural degrees of freedom of an  $A_2BX_4$  compound in search of a minimum of a given energy functional. In the inductive approach one offers a guess for the crystal structure of a given compound by analogy with the known structures of other compounds.

Most previous deductive approaches have focused on comparing a piece of the total (electronic + ionic) energy of different structures. The classical approach of crystal field stabilization energy (CFSE)<sup>[17]</sup> attempts to correlate the type of the observed local atomic structural motif (octahedral vs. tetrahedral) with the excess orbital energies by the nonspherical crystal field. This approach is applicable only to the cases where A or B is open shell transition metal. Even for this restriction the method was typically applied only to a subset of the known cases that do contain open shell A or B atoms, for example, the 44 cases in Reference [17]. The predictive power of the method is rather low: 46% success. Similarly, the approach of comparing point-ion Madelung energies of different structures was tested only for a small (18 compounds<sup>The A</sup>



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Table 1. Crystal structure types of A<sub>2</sub>BX<sub>4</sub> compounds. The labels b1–b38 and d1, d3, and d9 of structures in the first column are taken from Wyckoff,<sup>[5]</sup> whereas labels S1–S3 indicate  $Y_2CrS_4$ -type,  $Yb_3S_4$ -type, and  $Sr_2PbO_4$ -type structures, respectively. The notation "none" refers to cases where no known Pearson symbol nor mineral name exists.

Label	Prototype Compd.	Space Group	Pearson's Symbol/Mineral Name	No. of Compd.
b5	Al <sub>2</sub> MgO <sub>4</sub>	$Fd\overline{3}m(\Omega^7)$	cF56:Spinel	255
d9	Th <sub>3</sub> P <sub>4</sub>	I <del>4</del> 3d(T <sup>4</sup> )	none	87
b9	Fe₂CaO₄	$Pnma(D_{2b}^{16})$	none	78
b11	K <sub>2</sub> SO <sub>4</sub>	$Pnma(D_{2b}^{16})$	none	69
d3	Cr <sub>3</sub> S <sub>4</sub>	$C2/m(C_{2h}^{2h})$	mC14	57
b10	Al <sub>2</sub> BeO <sub>4</sub>	$Pnma(D_{2b}^{16})$	Olivine	48
b1	K <sub>2</sub> MgF <sub>4</sub>	$I4/mmm(D_{4b}^{17})$	none	41
b6	Mn <sub>3</sub> O <sub>4</sub>	$I4_1/amd(D_{4b}^{10})$	tl28;Hausmanite; distorted Spinel	27
b4	Ag₂HgI₄	$P\overline{4}2m(D_{2d}^1)$	tl14;Thiogallate	24
b33	Li <sub>2</sub> WO <sub>4</sub>	$R\overline{3}(C_{3i}^2)$	Phenakite	14
S1	Y <sub>2</sub> CrS <sub>4</sub>	$Pca2_1(C_{2v}^5)$	none	14
S2	Yb <sub>3</sub> S <sub>4</sub>	$Pnma(D_{2b}^{16})$	none	13
d1	Pb <sub>3</sub> O <sub>4</sub>	$P4_2/mbc(D_{4b}^{13})$	none	9
b21	Al <sub>2</sub> BaO <sub>4</sub>	P6 <sub>3</sub> 22(D <sub>6</sub> <sup>6</sup> )	none	7
S3	Sr <sub>2</sub> PbO <sub>4</sub>	$Pbam(D_{2h}^{9})$	none	6
b18	Na <sub>2</sub> SO <sub>4</sub>	Fddd(D <sup>24</sup> <sub>2b</sub> )	Thenardite	4
b2	K <sub>2</sub> PtCl <sub>4</sub>	$P4/mmm(D_{4h}^1)$	none	3
b3	$K_2Pd(NO_2)_4$	$P2_1/c(C_{2b}^5)$	none	2
b7	Cr <sub>2</sub> CuO <sub>4</sub>	$I\overline{4}2d(D_{2d}^{12})$	distorted Spinel	2
b13	KHSO <sub>4</sub>	$Pbca(D_{2b}^{15})$	none	2
b20	LiKSO <sub>4</sub>	$P6_3(C_6^6)$	none	2
b22	KNaSO <sub>4</sub>	$P\overline{3}m1(D_{3d}^3)$	Aphthitalite	2

Table 2. Crystal structure types of A

competing structures from empirically parametrized interatomic potential functions. This approach too, relies on selecting certain pieces of the full ion + electron total energy and on specific, approximate analytic forms of the potential terms. This approach has been applied<sup>[20]</sup> to a very restricted number of cases (54 compounds), predicting correctly 45 compounds (or 83% success).

Instead of minimizing pre-selected pieces of the full totalenergy, one might of course attempt to minimize an all-inclusive total energy expression, such as the density-functional and Hartree–Fock expressions. The density-functional calculations of the total energy were done by Marinelli et al. for In<sub>2</sub>MgS<sub>4</sub>,<sup>[16]</sup> Warren et al. and Thibaudeau et al. for Al<sub>2</sub>MgO<sub>4</sub>,<sup>[21]</sup> and Wei et al. for 18 spinel compounds.<sup>[22]</sup> The Hartree–Fock calculations of the total energy were done by Catti et al. for M<sub>2</sub>CrO<sub>4</sub> (M = Mg, Mn, Zn) and Mitchell et al. for M<sub>2</sub>ZnO<sub>4</sub> (M = Al, Fe).<sup>[23]</sup> Seko et al. used the cluster expansion method combined with density-functional calculations to investigate the ground state structures of 6 spinel compounds.<sup>[24]</sup> This successful deductive approach, however, is not easily applicable to large databases, works on a case by case basis, and requires optimization of the total energy with respect to  $\frac{1}{4}$ 

the conflict with the orbital radii approach is also not necessarily a contradiction. Third, ten cases are true errors.  $Cd_2PbO_4$ ,  $Cd_2PtO_4$ ,  $Cd_2SnO_4$ ,  $Ga_2PbO_4$ ,  $Pb_3O_4$ ,  $Tl_2CrO_4$ ,  $Sc_2MgO_4$ , and  $Ti_2NaO_4$  were determined experimentally to exist in S3,<sup>[6]</sup>S3,<sup>[6]</sup>S3,<sup>[13]</sup>b21,<sup>[5]</sup>d1,<sup>[5]</sup>b11,<sup>[5]</sup>b9,<sup>[12]</sup> and b9<sup>[12]</sup> structures (see Supporting Information), respectively, but appear in the map in the spinel (b5) region. As<sub>2</sub>NiO<sub>4</sub> has the d1 structure,<sup>[5]</sup> but is placed in the map in the region of b33. Ni<sub>2</sub>RhS<sub>4</sub> is reported to exist in the spinel structure,<sup>[6]</sup> while its orbital radii place it inside the d3 region. The spinel and d3 structures (see Figs. S1 and S5 in the Supporting Information) are closely related: both have their 2/3 cations octahedrally coordinated; their anion sub-lattices (fcc in6lb-308.2(and)-308.9hcapinboth

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2

2

A and B ions,<sup>[37]</sup> including every compound that can be included. There are 40 compounds that are included in orbital radii maps but cannot be included in Shannon's radii maps (see Supporting Information). The 5 compounds containing U (not included in orbital radii maps) are included in Shannon's radii maps. The success rates of the structure-field maps and cation-distribution map using Shannon's crystal radii<sup>[37]</sup> (see Figs. S10 and S11 in the Supporting Information) are 92% and 74% compared to the success rates of the orbital radii maps at 98% and 96%, respectively. Shannon's radii



